

Li⁺ ION CONDUCTION IN COBALTOUS COBALTICYANIDE DOPED WITH LITHIUM CHLORIDE

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Abstract : Cobaltous cobalticyanide (a Prussian Blue type of solid with large interstitial cavities) doped with lithium chloride is found to exhibit Li⁺ ion conduction. The data on temperature variation of conductivity at different concentrations of LiCl are presented.

1. Introduction

Prussian Blue (Fe₄(Fe(CN)₆)₃) and related heavy metal hexacyanides form a class of crystalline solids with complex ions whose structure is well understood.^{1,2,3} They have similar face-centered cubic arrangement of metal cations at the corners of unit cubes linked by cyanide ions placed along the edges. A peculiar property arising from this structure is that the unit cells are unusually large (lattice constant $\sim 10 \text{ \AA}$).^{1,2,3} As the result the crystal can accommodate foreign molecules and ions as interstitial impurities.^{1,2,3} We have noted that Prussian Blue type compounds doped with Li salts exhibit ionic conduction. Electronic conductivity (30°C) of Prussian Blue is $\sim 3 \times 10^{-6} \Omega^{-1} \text{ m}^{-1}$. However, it was found that compound with identical structure cobaltous cobalticyanide (Co₄(Co(CN)₆)₃) has smaller electronic conductivity (30°C) $\sim 4 \times 10^{-8}$ and high stability towards thermal degradation. Consequently this material is more suitable for studying Li⁺ ion transport in metal hexacyanides doped with lithium salts. In this note we report our observations on Li⁺ ion conduction in Co₄(Co(CN)₆)₃ doped with LiCl.

2. Experimental

$\text{Co}_4(\text{Co}(\text{CN})_6)_3$ was prepared by adding potassium cobalticyanide (Aldrich) solution (0.1 M) dropwise to a solution of cobalt nitrate (0.5 M). (Cobalt nitrate kept in excess to avoid formation of double salts containing potassium). The pink precipitate of $\text{Co}_4(\text{Co}(\text{CN})_6)_3$ separated by filtration was washed with distilled water until the filtrate is free from potassium. The powder was dried in vacuum at 140°C for several hours to remove water of hydration. (Anhydrous material has a deep blue colour). The doping with LiCl was done by the following method. $\text{Co}_4(\text{Co}(\text{CN})_6)_3$ was mixed with the desired amount of LiCl, the mixture homogenized and then dried in vacuum at 140°C to remove all moisture. The dried powder was compacted between carbon electrodes in a glass tube (diameter ~ 0.6 cm) to a pressure of 800 psi until a pellet (length ~ 0.5 cm) was formed. Ends of the tube were sealed with epoxy resin, the sample immersed in a thermostatic oil bath and a.c. (40 Hz) conductivity measured. (Modified Electronic Instruments Conductivity Bridge Model MC - 1, operated at 9 V). The d.c. conductivity was also measured by the polarization (blocking electrode) method^{4,5} and found to be of the same order as the a.c. values. The rapid decrease in conductivity with time approaching a limit comparable to intrinsic electronic conductivity of $\text{Co}_4(\text{Co}(\text{CN})_6)_3$ clearly demonstrated that the charges carried are ionic.

3. Results and Discussion

Figure 1 gives a plot of $\ln \sigma$ vs T^{-1} for different concentrations of LiCl (c , measured as a percentage by wt). In each case the graph is a straight line showing that the relation,

$$\sigma = \sigma_0 e^{-E/kT} \quad (1)$$

is satisfied, both E and σ_0 are found to depend on the degree of doping (ie, c). The plots of E vs c and σ_0 vs c are shown in Figures 2 and 3. The conductivity was found to be maximum ($\sigma_{30} \simeq 2.5 \times 10^{-3} \Omega^{-1}\text{m}^{-1}$) when $c \simeq 34\%$ wt (Figure 4) and minimum value of E also corresponds to this value of c . Again the minimum value of σ_0 happens occur when the level of doping is $\sim 34\%$. It is possible that the critical point occurs when the interstitial cavities are nearly filled with LiCl. A simple calculation based on estimate of the volume of an interstitial cavity (using following data: lattice constant, ionic radii of Co^{2+} , Co^{3+} and CN^-) and density of solid LiCl indicate that the cavities get completely filled when $c \simeq 41\%$. The Li^+ ion mobility probably results from ionization of LiCl into Li^+ and Cl^- ions by the crystal field of $\text{Co}_4(\text{Co}(\text{CN})_6)_3$. The smaller Li^+ ion become mobile and move through the interstices. Cl^- ions could also have some mobility, but we did not succeed in detecting this experimentally.

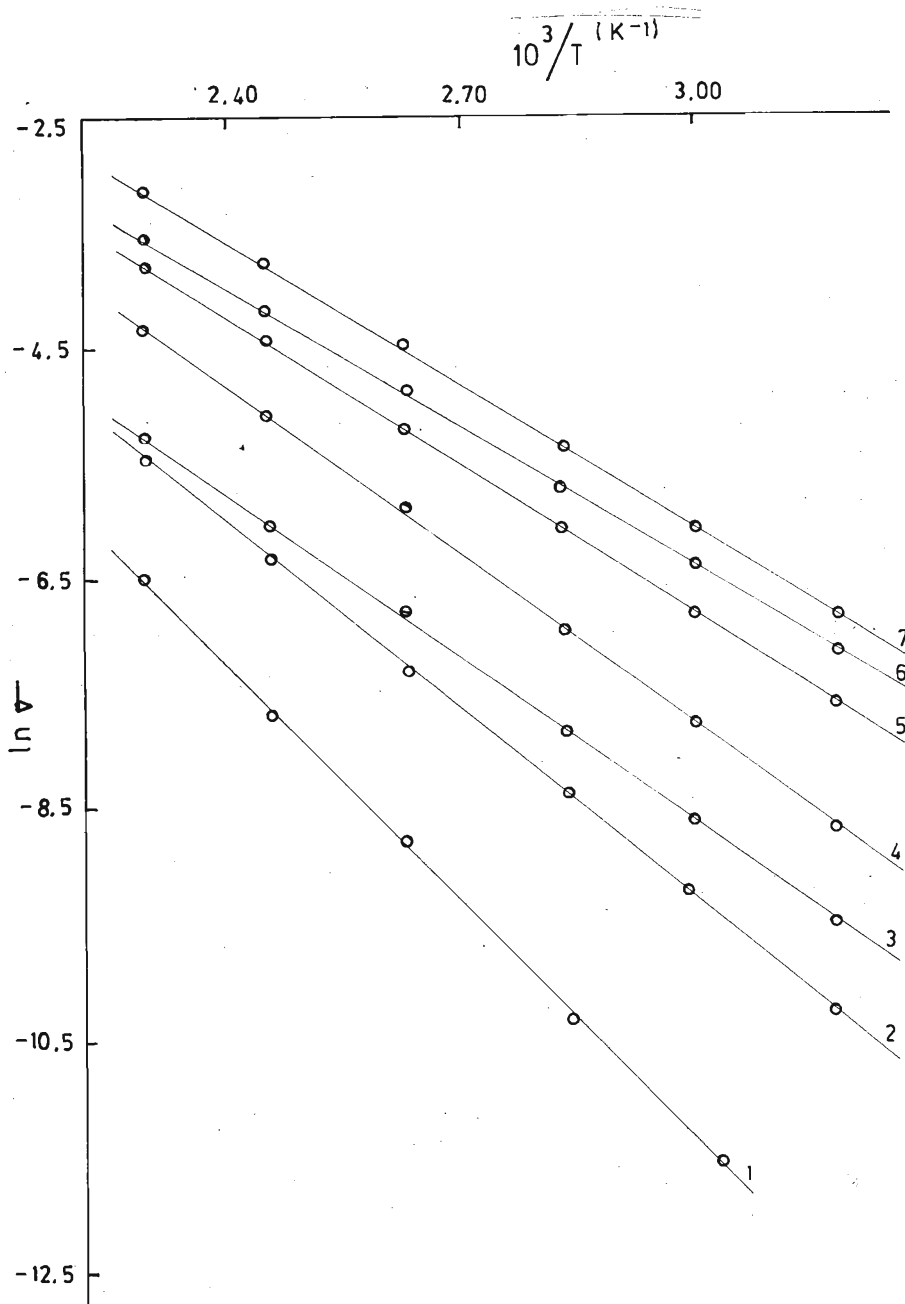


Figure 1. Plot of $\ln \sigma_0$ vs. T^{-1} (σ in $\Omega^{-1} \text{ m}^{-1}$). Level of deoping (ie, % of LiCl by wt).

- | | | | |
|----------|----------|----------|----------|
| (1) 50 | (2) 16.7 | (3) 41.2 | |
| (4) 23.1 | (5) 37.5 | (6) 28.6 | (7) 41.2 |

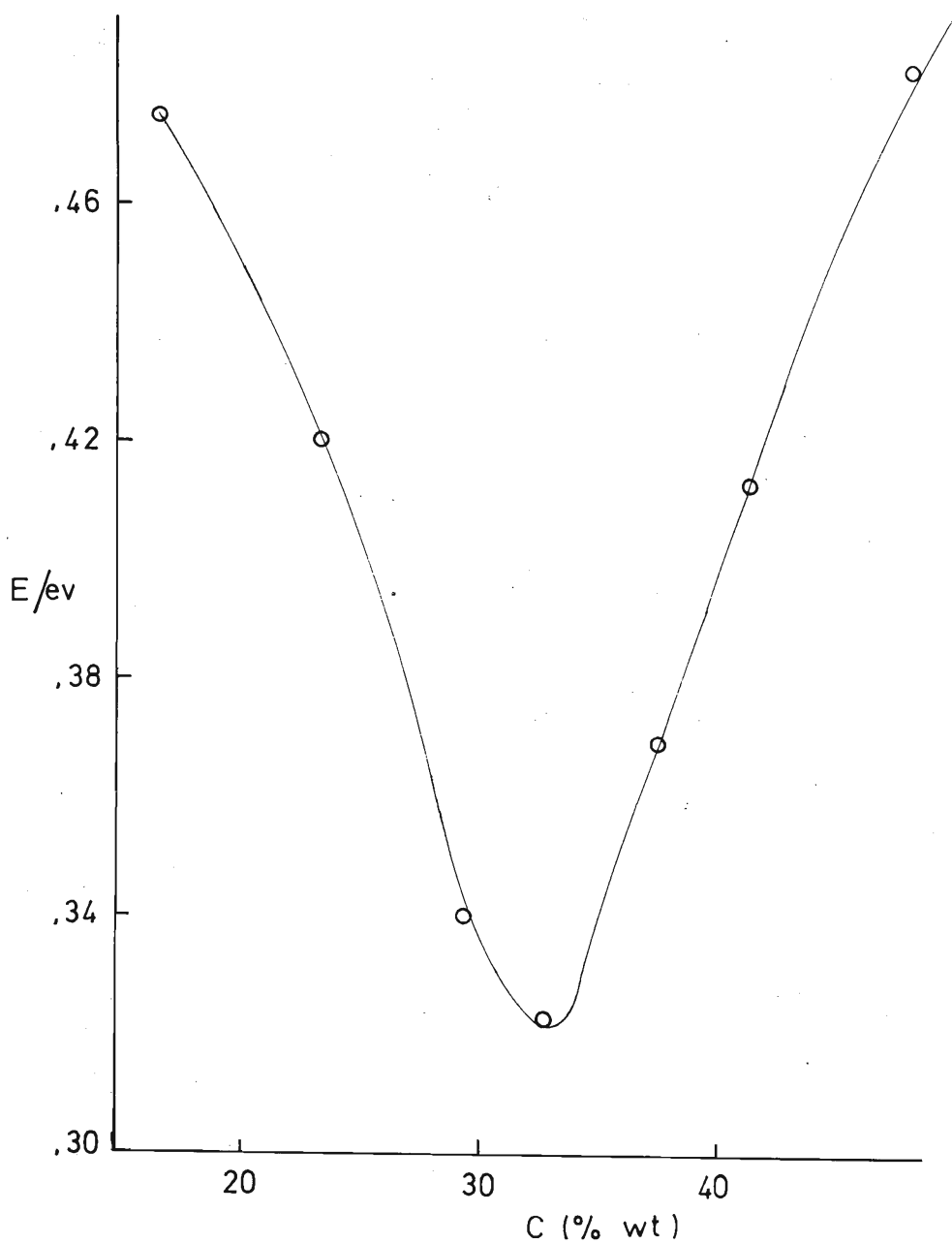


Figure 2. Plot of E vs C.

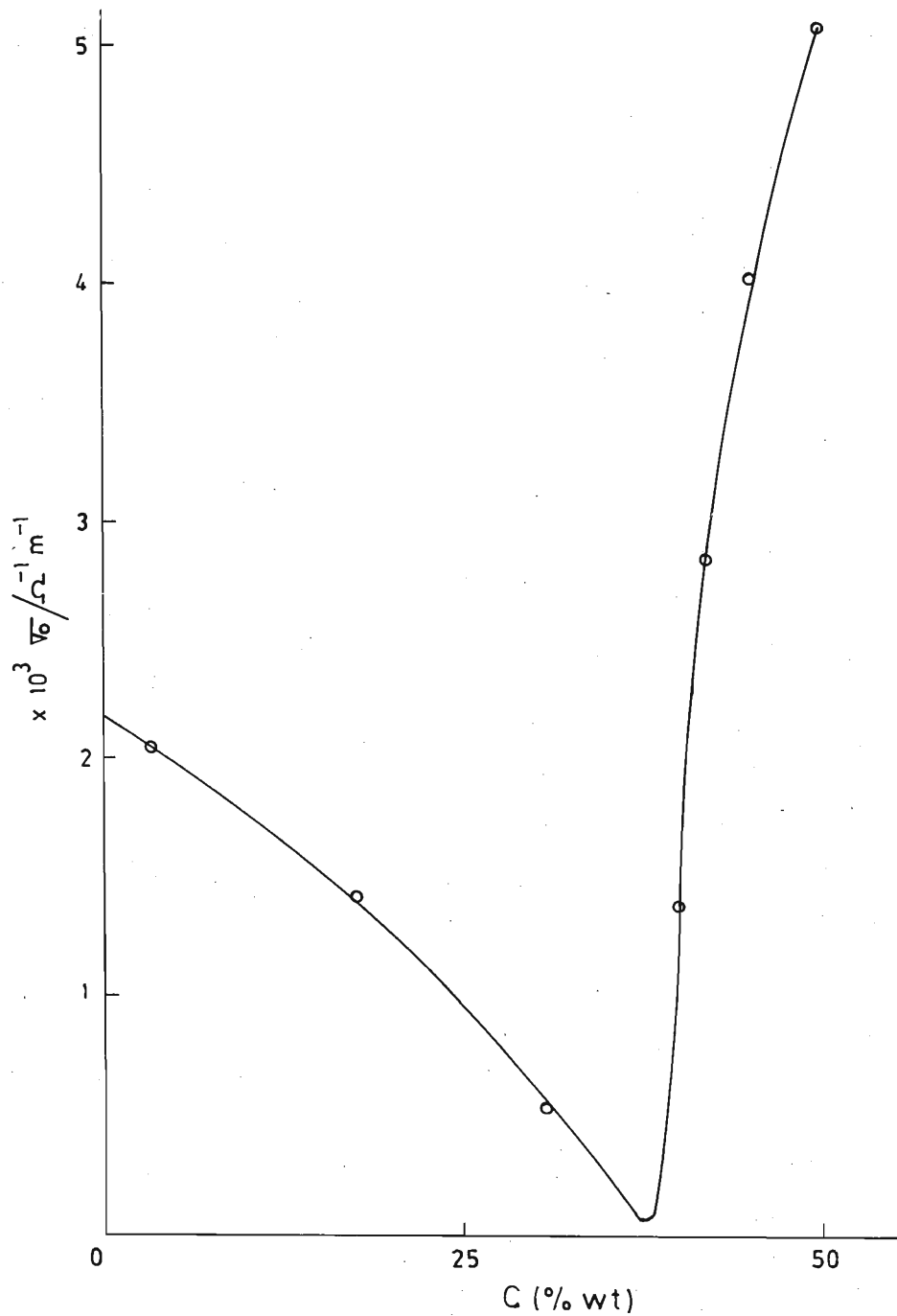


Figure 3. Plot of σ_0 ($\Omega^{-1} m^{-1}$) vs c .

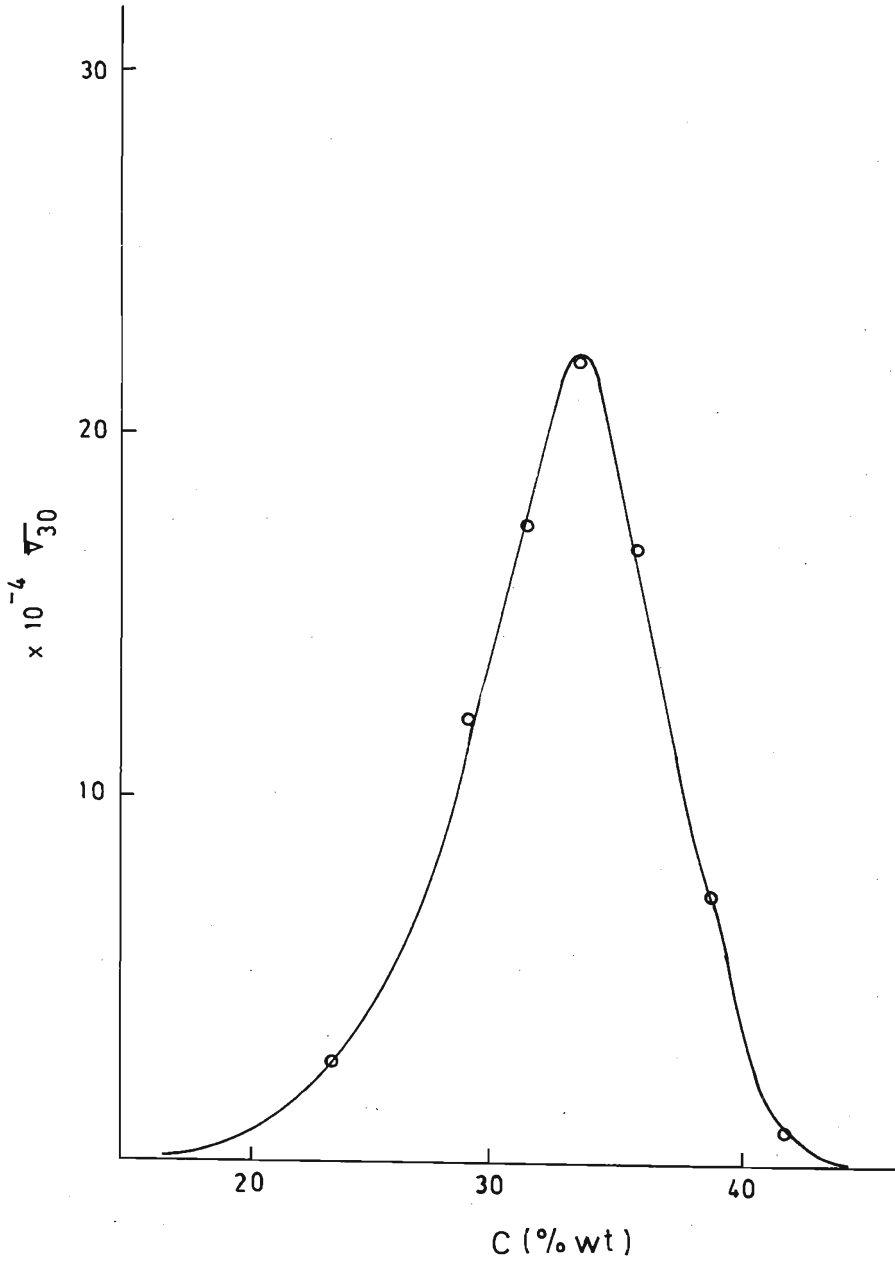


Figure 4. Plot of σ_{30} ($\Omega^{-1}\text{m}^{-1}$) vs c.

4. Conclusion

Further experimental work is necessary to elucidate the exact nature of Li⁺ ion transport in this material. As interstitial cavities are large, it is very likely that Cl⁻ ions also have some mobility in this material. Measurement of Cl⁻ mobility would assist in understanding the mechanism of ionic conduction in Prussian Blue type materials. Unfortunately we did not have facilities for determination of the mobilities of Li⁺ and Cl⁻ ions separately.

References

1. AYERS, J. B. & WAGGONERS, W. H., (1971) *J. Inorg. Nucl. Chem.* **33**: 721.
2. BONNETTE, A. K. & ALLEN, J. F., (1971) *Inorg. Chem.* **10**: 1613.
3. GANGULI, S. & BHATTACHARYA, M., (1983) *J. Chem. Soc. Faraday Trans. 1*, **79**: 1513.
4. KENNEDY, J. H. (1977) *Physics of solid electrolytes*, ed. S. Geller (Springer, New York).
5. WAGNER, J. B. & WAGNER, C. (1957) *J. Chem. Phys.* **26**: 1597.